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Electrical Characterization of Tin Oxide Thin Films Prepared by Dip Coating Technique

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ABSTRACT: The influence of bath temperature on tin oxide Nano layers on the surface of glass substrate bysol-gel dip coating method was investigated. The bath temperature was varied from 313 K to 373 K in steps of 20 K. The deposited films were annealed at 673 K. These films were characterized by dc conductivity measurements. The dc electrical conductivity was measured in the temperature rangeof 303-403 K and was found to increase with temperature. The results indicated that the material could be used for solar and laser applications.

KEYWORDS: tin oxide thin films, dc conductivity, solar and laser applications.

I. INTRODUCTION

Transparent conducting tin oxide (SnO_2) semiconductors are widely used as transparent films in electronic devices such as thin filmsolar cells, display devices, electro-chromic display, defogging aircraft and automobile windows, gas sensors, light emitting diodes, flat-panel display and other optoelectronic devices, etc., [1-3]. Tin oxide has been prepared as a promising alternative anode material for micro batteries. Recently, many researchers have reported the preparation and properties of SnO_2 films such as chemical precipitation method [4], spray pyrolysis [5], sol-gel dip coating [6], plasma oxidation [7], etc.

Even though several methods have been employed to form a transparent SnO_2 thin film, to the best of the knowledge of the authors, this is the first attempt to prepare SnO_2 thin filmsannealed at low temperature. Based on these aspects, the authorshave presented the work on the growth of SnO_2 films by a cost effective dip technique which becomes very important for their practical applications.

II. EXPERIMENTAL PROCEDURE

The coating solution used for the experiment was prepared from anhydrous $SnCl_4$ by process. Pure SnO_2 sols were prepared from $SnCl_4$, distilled water, n-propanol (C_3H_7OH) and isopropanol (2- C_3H_7OH) at a molar ratio $SnCl_4:H_2O: C_3H_7OH:2- C_3H_7OH = 1:9:9:6.$ At first, the $Sncl_4$ was dropped into two-thirds of the total amount of propanol. One-third of the total distilled water dissolved in the remaining n-propanol (C_3H_7OH) was added for prehydrolysis of the tin precursor. It was stirred for 1hr and after that the previously prepared sol was mixed with the solution of the remaining amount of H_2O dissolved in the prescribed amount of isopropanol (2- C_3H_7OH). It was again stirred for 1hr. With this procedure, clear and homogeneous sols were prepared. The bath temperature varied from 313 to 373 K in steps of 20 K. The prepared films were annealed at 673 K.

III. RESULTS AND DISCUSSION

The electrical resistivity of the films was measured using a four probe setup in the temperature range of 303-403 K as shown in Fig 1. The electrical measurement was performed by making four electrical contacts with sample surface. Two of the probes were used to measure the current and the other two probes were used to measure the corresponding voltage.

The electrical conductivity of the samples was calculated from the following equation,

$$\rho = \frac{\pi}{\ln(2)} \times \frac{V}{I} \times t \quad \frac{----(1)}{\text{DOI:}10.15680/\text{JJIRSET.}2016.0508076}$$

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Where (I)is the current, (V) is the voltage and (t) is the thickness of the sample. In order to investigate the rectifying behaviour of the samples at different temperatures, fourprobe setup was connected to the Keithley 6517B electrometer by applying voltage of 1 to 10 V in steps of 1V. After the application of each volt increment, current was automatically measured with a delay time of 2 sec. I-V data at each selected temperature was transferred to the computer and stored for further analysis. The current through the sample as function of applied voltage was measured in the temperature range of 303-403 K. I-V measurements for each sample were carried out by reversing the polarity of the contacts several times.







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I-V characteristics of SnO₂ thin films prepared at different bath temperature of 313-373Kwere annealed at temperature 673 K as shown in fig. 1 (a-f)313-373 K.The curves are found to be linear, indicating ohmic conduction due to the thermally generated carriers [8].

In the ohmic region, the electrical conduction is governed only by the resistance of the films and controlled by thermally activated carriers. This power continues until the injected free carrier density becomes comparable with thermally created carrier density and at sufficiently higher fields the conduction current is mainly due to the injected space charge [9].

It is observed that for the same applied filed, the current increases with increasing temperature in ohmic region for all samples. As the temperature increases, the probability of thermal ionization of the trapping centres increases too. Hence a shift in the quasi Fermi level occurs and this causes the lowering of the barrier across which the electrons are transported and the conduction becomes nearly ohmic [10].

The temperature dependence of DC conductivity (σ_{dc}) for SnO₂ thin films prepared at different bath temperatures (313-373 K) were annealed temperature 400°C as shown in fig 2. In all cases conductivity increased with the increase in temperature for the entire range of 303 to 403 K. This is the characteristic of semiconducting behaviour.

--(2)



 $\rho = \rho_a \exp(E_a/KT)$

Fig 2 Temperature Vs Conductivity of tin oxide thin films at different bath temperatures







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The activation energy (E_a) determined, at different constant applied voltages using the relation $\rho = \rho \rho \exp(-E_a/KT)$, by plotting $\ln \sigma$ against 1000/T (fig 3). Jonscher and Anasari suggested that ∇E values less than 0.8 eV would indicate that the dominant charge transport is through the motion of electrons, as where ∇E values higher than 0.8 eV reflect that the dominant charge transport is ionic. Therefore the values of ∇E for all samples indicate that the electronic conduction is dominated in these samples, which provides support to the proposed analysis[11-12].

The activation energies for the charge transport mechanics were calculated using he initial rise method of Perlman [13] by isolating the individual peck from the composite curves. The slopes of log 1 Vs 1/T plots from the initial rising portion of the individual pecks were used to calculate the activation energies as shown in fig 3.

The DC conductivity of SnO_2 thin film prepared at bath temperature (313 K) was $0.2327 \times 10-3$ S/cm. As the bath temperature increased the conductivity was found to change from $0.2327 \times 10-3$ S/cm. After the bath temperature increase, the conductivity was found to change from $0.2327 \times 10-3$ S/cm. The conductivity of the SnO_2 thin films increases slowly by a few orders of magnitude with the increase of bath temperature. The slight increase in the conductivity of SnO_2 thin films with the increase of bath temperature may be not only due to the mobility of the charge carriers at the Fermi level, but also due to the presence of high amount of conducting species in all the samples [14-16].





Fig3 1000/T Vslno of tin oxide thin filmsat different bath temperatures

The activation energy values for tin oxide thin films can be calculated from the slope of the Arrhenius plot over the entire temperature range (313-373 K). The calculated activation energy was found to be 0.875-0.116 eV. It is clear from the table that, as film temperature of the sample increased, conductivity also increased. The decrease in activation energy with temperature was shown inTable 1.



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Table:1 DC Conductivity, Activation Energy and different bath temperature of tin oxide thin films

Bath Temperature (K)	Conductivity $\sigma \times 10^{-3} \Omega$ -cm) ⁻¹	Activation energy, Eg(eV)
313	0.2227×10 ⁻³	0.800
333	0.2327×10	0.600
353	0.2902×10 ⁻³	0.161
373	0.5673×10 ⁻³	0.116

IV. CONCLUSION

The tin oxide (SnO_2) thin films were prepared by sol-gel at different bath temperatures annealed at 673 K. The linear I-V characteristics may be due to semiconducting nature of the films. The curves were found to be linear, indicating ohmic conduction due to the presence of thermally generated carriers. The estimated electrical conductivity of SnO₂ films was found to be in the range of 0.2327 x 10⁻³ (Ω -cm)at lower temperature (313K) and 0.5673 x 10⁻³ Ω -cm at higher temperature (373K). It is also observed that the conductivity of SnO₂ thin films increased with increasing bath temperature. It is not only due to the mobility of the charge carriers at the Fermi level, but also due to the presence of high amount of conductivity possess the lowest activation energy and the value is found to be in the semiconducting range.

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